

Piezoelectric Properties of (Li,Mn)-doped Ba(Zr,Ti)O₃-(Ba,Ca)TiO₃ Lead-free Piezoelectric Ceramics

Jiayi LIU^{1‡}, Yuka TAKAGI¹, and Hajime NAGATA¹
(¹Grad. School Electr. Eng.,Tokyo Univ. of Science)

Abstract

Li₂CO₃ and MnCO₃ co-doped (1-*x*)Ba(Zr_{0.2}Ti_{0.8})O₃-*x*(Ba_{0.7}Ca_{0.3})TiO₃ [BZT-*x*BCT] ceramics were prepared by conventional solid-state reaction method and their piezoelectric properties were examined in detail. The product of piezoelectric constant *d* and mechanical quality factor *Q_m* (*d* × *Q_m*) at MnCO₃ 0.6 wt% and Li₂CO₃ 1 wt% doped BZT-0.6BCT ceramics was approximately 4 times larger than that of non-doped BZT-BCT ceramics. In addition, BZT-BCT+0.6Mn+1.0Li showed relatively higher vibration velocity *v*_{0-p} at small applied electric field (*E*_{0-p} = 15 V/mm), which is related to the small signal *d*₃₁ × *Q_m*.

1. Introduction

High-power piezoelectric ceramic devices such as ultrasonic vibrator for ultrasonic cleaner, has been widely used¹⁾. For high-power piezoelectric applications, vibration velocity *v*_{0-p} is one of the important parameters, which is generally known as proportional to a product of piezoelectric constant *d* and mechanical quality factor *Q_m* (*d* × *Q_m*). Then, both larger *d* and *Q_m* values are required for the practical applications of high-power piezoelectric devices. At practical devices, hard-type Pb(Zr,Ti)O₃ [PZT] based ceramics are mainly used because of high *d* × *Q_m* value. However, the *v*_{0-p} of PZT shows non-linearity at large amplitude vibration, moreover, PZT contains harmful element PbO²⁾. Therefore, development of lead-free piezoelectric materials is required for even high-power piezoelectric devices.

(1-*x*)Ba(Zr_{0.2}Ti_{0.8})O₃-*x*(Ba_{0.7}Ca_{0.3})TiO₃ [BZT-*x*BCT] has been reported to have a relatively high piezoelectric constant *d*, making it a potential candidate for lead-free piezoelectric ceramics³⁻⁸⁾. This is because BZT-*x*BCT has a morphotropic phase boundary (MPB) starting from a tricritical triple point in which it has large piezoelectric constant *d*₃₃ about 620 pC/N⁹⁾. In this study, we selected the composition of *x* = 0.6 where *T_C* is at around 100°C and the piezoelectric characteristics are relatively stable for temperature¹⁰⁾. However, the *d* of this composition is relatively low as compared with that of MPB composition and the *Q_m* is not large enough for practical high-power piezoelectric application. To solve these problems, we focused on the doping effect of Mn ions as hard dopant and Li ions for liquid-phase sintering. Therefore, doping effect of Li and Mn ions for BZT-0.6BCT ceramics was

examined in detail to improve piezoelectric properties such as *v*_{0-p} and *d* × *Q_m*.

2. Experimental procedure

BZT-0.6BCT + Li₂CO₃ *y* wt% [BZT-0.6BCT+Li_{*y*}] and BZT-0.6BCT + MnCO₃ 0.6 wt% + Li₂O₃ *y* wt% [BZT-0.6BCT+0.6Mn+Li_{*y*}] (*y* = 0, 0.4, 1.0, 1.5) ceramics were prepared by conventional solid-state reaction method. Carbonate and oxide powders, i.e., BaCO₃, TiO₂, ZrO₂, CaCO₃, Li₂CO₃, and MnCO₃ (≥ 99.9%), were used as starting materials. These starting raw materials without Li₂CO₃ were weighed in accordance with the chemical formula 0.4Ba(Zr_{0.2}Ti_{0.8})O₃-0.6(Ba_{0.7}Ca_{0.3})TiO₃ and then ball-milled with zirconia balls in ethanol for 5 h. The mixture was dried and uniaxially pressed into pellets, then calcined at 1000°C for 4 h. The calcined pellets were ground and ball-milled for 15 h with Li₂CO₃ addition. After drying, the mixed powders were press into pellets and subjected to a cold isostatic pressing (CIP) treatment at 150 MPa, then sintered at 1300°C and 1400°C for 2 h.

The phases of the sintered samples were identified by X-ray diffraction. Scanning electron microscopy (SEM) was used to observe the microstructure. For the measurement of the piezoelectric properties, the samples were cut into appropriated shapes of the (31) mode (12×3×1 mm³) and silver-electrode on both surfaces, then poled in a silicone oil bath by applying dc electric fields of 3.5 kV/mm for 20 min at room temperature. The piezoelectric properties were investigated by a resonance-antiresonance method using an impedance analyzer (HP4294A) and the vibration velocity *v*_{0-p} was measured using a laser doppler vibrometer (ONOSOKKI LV1710).

3. Results and discussion

All (Li, Mn)-doped BZT-BCT ceramics showed high relative density ratio over 94% of the theoretical density and a single-phase perovskite structure was observed in each sample from XRD patterns. **Fig. 1(a)** and **Fig. 1(b)** show small signal *d*₃₁ and *Q_m* as a function of Li₂CO₃ content of BZT-BCT+Li_{*y*} and BZT-BCT+0.6Mn+Li_{*y*} ceramics. The *d*₃₁ values were improved by Li₂CO₃ doping and the maximum obtained at *y* = 1 wt% under all experimental conditions. Among them, *d*₃₁ value on BZT-BCT+1.0Li showed largest about 160 pC/N. This is because Li₂CO₃ doping improved dielectric constant ϵ_r which is responsible for the increase of *d*₃₁. Also,

the Q_m values were enhanced by $MnCO_3$ doping. As shown in Fig. 1(b), the Q_m value was approximately 2 times larger than that of non-Mn doped ceramics at $y = 1$ wt%. This is because Mn ions work as acceptor that increase oxygen vacancies. Therefore, the Q_m increased and piezoelectric characteristics became harder due to the domain pinning effect.

Fig. 2 shows $d_{31} \times Q_m$ as a function of Li_2CO_3 content of BZT-BCT+Li_y and BZT-BCT+0.6Mn+Li_y ceramics. The value of $d_{31} \times Q_m$ is the highest at BZT-BCT+0.6Mn+1.0Li ceramic. This is because the Q_m has large enhancement due to the co-doping

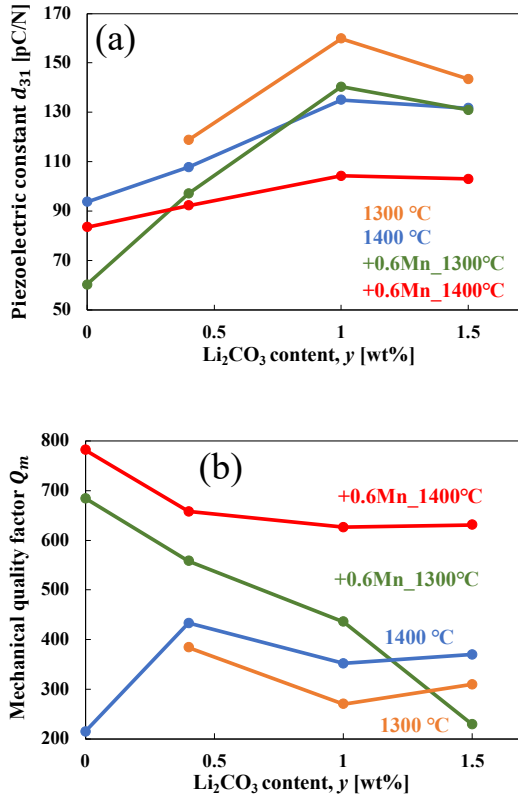


Fig. 1 (a) d_{31} and (b) Q_m as a function of Li_2CO_3 content of BZT-BCT+Li_y and BZT-BCT+0.6Mn+Li_y ceramics.

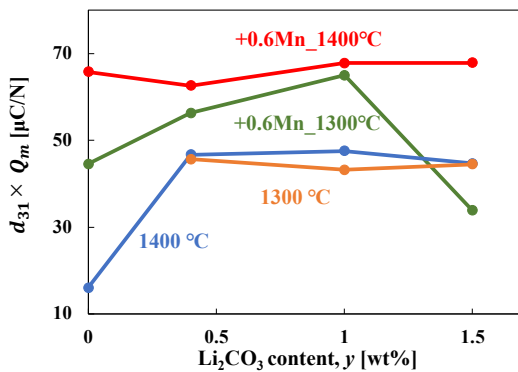


Fig. 2 $d_{31} \times Q_m$ as a function of Li_2CO_3 content of BZT-BCT+Li_y and BZT-BCT+0.6Mn+Li_y ceramics.

effect of Li_2CO_3 and $MnCO_3$, as compared with the small deterioration of d_{31} value. As a result, the $d_{31} \times Q_m$ value was approximately 4 times larger than that of non-doped BZT-0.6BCT ceramics. Also, BZT-BCT+0.6Mn+1.0Li showed the highest v_{0-p} at small applied electric field ($E_{0-p} = 15$ V/mm), which is related to the small signal $d_{31} \times Q_m$. From these results, (Li, Mn)-doped BZT-0.6BCT has excellent piezoelectric properties, which could be one of the potential candidates for lead-free materials on high-power piezoelectric applications.

4. Conclusion

In this work, Li_2CO_3 and $MnCO_3$ co-doped BZT-0.6BCT ceramics were prepared by conventional solid-state reaction method. The piezoelectric properties have been investigated, and as a result, the $d_{31} \times Q_m$ value was approximately 4 times larger than that of non-doped BZT-BCT ceramics. In addition, BZT-BCT+0.6Mn+1.0Li showed relatively higher v_{0-p} at small applied electric field ($E_{0-p} = 15$ V/mm), which is related to the small signal $d_{31} \times Q_m$. From these results, (Li, Mn)-doped BZT-0.6BCT has excellent piezoelectric properties which could be one of the potential candidates for lead-free ceramics on high-power piezoelectric applications.

5. References

1. X. Yan, K. H. Lam, X. Li, R. Chen, W. Ren, X. Ren, Q. Zhou, and K. K. Shung, *IEEE Trans. Ultrason. Ferroelectr. Freq. Control*, **60**, 1272-1276 (2013).
2. H. I. Humburg, M. Acosta, W. Jo, K. G. Webber, and J. Rödel, *J. Eur. Ceram. Soc.*, **35**, 1209-1217 (2015).
3. M. Zakhovzheva, L. A. Schmitt, M. Acosta, W. Jo, J. Rödel, and H.-J. Kleebe, *Appl. Phys. Lett.*, **105**, 112904 (2014).
4. M. Sanlialp, V. V. Shvartsman, M. Acosta, B. Dkhil, and D. C. Lupascu, *Appl. Phys. Lett.*, **106**, 062901 (2015).
5. M. Acosta, N. Novak, G. A. Rossetti Jr., and J. Rödel, *Appl. Phys. Lett.*, **107**, 142906 (2015).
6. D. R. J. Brandt, M. Acosta, J. Koruza, and K. G. Webber, *J. Appl. Phys.*, **115**, 204107 (2014).
7. M. Zakhovzheva, L. A. Schmitt, M. Acosta, H. Guo, W. Jo, R. Schierholz, H.-J. Kleebe, and X. Tan, *Phys. Rev. Appl.*, **3**, 064018 (2015).
8. S. Zhukov, Y. A. Genenko, M. Acosta, H. Humburg, W. Jo, J. Rödel, and H. Seggern, *Appl. Phys. Lett.*, **103**, 152904 (2013).
9. Liu W, Ren X, *Phys Rev Lett* **103**:257602 (2009)
10. H. Acosta, N. Khakpash, T. Someya, N. Novak, W. Jo, H. Nagata, G. A. Rossetti, Jr., and J. Rödel, *PHYS. REV. B*, **91**, 104108 (2015).